Arthropods face several key challenges in processing concentrated feedstocks of proteins (silk dope) into solid, semi-crystalline silk fibres. Strikingly, independently evolved lineages of silk-producing organisms have converged on the use of liquid crystal intermediates (mesophases) to reduce the viscosity of silk dope and assist the formation of supramolecular structure. However, the exact nature of the liquid-crystal-forming-units (mesogens) in silk dope, and the relationship between liquid crystallinity, protein structure and silk processing is yet to be fully elucidated. In this review, we focus on emerging differences in this area between the canonical silks containing extended-β-sheets made by silkworms and spiders, and ‘non-canonical’ silks made by other insect taxa in which the final crystallites are coiled-coils, collagen helices or cross-β-sheets. We compared the amino acid sequences and processing of natural, regenerated and recombinant silk proteins, finding that canonical and non-canonical silk proteins show marked differences in length, architecture, amino acid content and protein folding. Canonical silk proteins are long, flexible in solution and amphipathic; these features allow them both to form large, micelle-like mesogens in solution, and to transition to a crystallite-containing form due to mechanical deformation near the liquid–solid transition. By contrast, non-canonical silk proteins are short and have rod or lath-like structures that are well suited to act both as mesogens and as crystallites without a major intervening phase transition. Given many non-canonical silk proteins can be produced at high yield in E. coli, and that mesophase formation is a versatile way to direct numerous kinds of supramolecular structure, further elucidation of the natural processing of non-canonical silk proteins may to lead to new developments in the production of advanced protein materials.

1. Introduction: silk spinning and liquid crystallinity
(a) Silk comprises multiple diverse materials produced by distinct groups of arthropods

Silk production occurs in a diverse range of arthropods including crustaceans, mites, centipedes and insects, and it is now recognized that silk production has evolved independently more than 23 times within the arthropods [1]. For the purposes of this review, we define silks as protein materials that are converted (spun) from a highly concentrated liquid feedstock (dope), and which undergo a liquid-to-solid phase transition concurrently with being mechanically drawn from the silk gland into the external air (pultrusion).

Silk materials are semi-crystalline polymers consisting of ordered domains containing crystallites with precise secondary structures and high H-bonding density, interspersed with disordered amorphous domains with a lower H-bonding density [2,3]. Of particular note, as it serves as a basis for our sub-classification, the structures of the ordered crystallites may be β-sheets, α-helical coiled-coils, collagen helices or the polyglycine II structure [1,4]. Further structural variation derives from crystallite orientation, a feature distinguishing the extended-β-sheet
Figure 1. Canonical and non-canonical silks. (a) Silkworm (*B. mori*) cocoon fibres, a canonical silk. (b–d) Non-canonical silks produced by other insect species. (b) Lacewing (*M. signata*) egg-stalk silk, photograph by Holly Trueman. (c) Sawfly (*Nematus oligospilus*) cocoon silk. (d) Honeybee (*Apis mellifera*) silk and wax on cell caps of a hive, photograph by Alex Wild.

structure (amide backbones parallel to fibre axis) from the cross-β-sheet structure (amide backbones perpendicular to fibre axis). The canonical silks such as the cocoon fibres of silkworms (*Bombyx mori*; Lepidoptera) and dragline silks of orb-spiders (*Araneidae*) contain crystallites of the extended-β-sheet type; an adaptation that has also arisen independently in wasps (Hymenoptera) [4]. Silks containing crystallites of other types, referred to here collectively as non-canonical silks, are produced by at least nine groups of insects [1,4]. In this review, we compare canonical silks made by the silkworm and orb-spiders with the non-canonical silks made by three insect taxa (figure 1): the aculeates (ants, bees and wasps; Hymenoptera: Aproctida), sawflies of the tribe Nematini (*Hymenoptera: Tenthehridinae*) and lacewings (*Neuroptera: Chrysopidae*).

X-ray scattering experiments on aculate silk reveal longitudinal d-spacings of 0.51 nm, indicating that crystallites are of the coiled-coil type and orientated parallel to the fibre axis, and lateral spacings of approximately 2.5 nm corresponding to lateral packing between superhelices [4,6]. These data suggest superhelices share a common orientation but not ordered in a lattice-like way with respect to adjacent superhelices. Hence, the ‘crystallites’ in aculate silk are probably best understood as single-coiled-coil superhelices. Sawflies in the tribe Nematini make pupal cocoons in which another superhelical structure, the collagen triple helix, is oriented on average parallel to the fibre axis [4]. The third non-canonical silk we consider in this review are egg-stalks made by mother green lacewings (*M. signata*), which have the cross-β-sheet structure [7].

Understanding the relationship between silk protein amino acid sequence, processing and the hierarchical structure of the final material spans the fields of physiology, protein folding, molecular evolution and materials science. We will argue that there are compelling reasons to believe that the process of spinning canonical and non-canonical silks is substantially different, constituting two separate general strategies and hence ‘fitness peaks’ [8] in the adaptive landscape of the system as a whole—including protein structure and length, gland morphology and the material’s mechanical properties before, during and after spinning. Understanding the full range of silk processing modes used by organisms is invaluable in guiding the endeavours for a long-standing biotechnological challenge: true biomimetic manufacture of silks and faithful replication of a silk fibre’s mechanical properties [9].

(b) Principles of silk spinning by arthropods

While the key challenges of silk processing are shared across disparate silk-producing taxa, we note that common challenges may be overcome via disparate strategies. We summarise the main challenges in silk processing under three headings.

(i) Overcoming flow viscosity

To facilitate transition to a solid material, silk dope is a highly concentrated solution of proteins, frequently 20–40% of dry weight [10]. Initial rheological measurements on freshly extracted spider and silkworm dopes found both to be viscoelastic materials, similar to a high molecular weight polymer acting as a weak gel, with a high viscosity in the KPa s range [10–12]. Taken in isolation and from a polymer processing perspective, this material would present these animals with a considerable challenge to process into a micrometre-sized fibre as the forces required to flow the material down the fibre duct are prohibitively high [13]. However, microscopic analysis of the silk spinning apparatus in both animals provided a solution; unspun canonical silks appear to be liquid crystals [14–16]. In the silk ducts of silkworm silk glands and orb-spider major ampullate glands, liquid crystalline textures indicating the presence of nematic mesophases (see §1c) are observed, which persist in the dope until disappearing shortly before the spinneret [14]. The presence of liquid crystallinity in silk dopes reduces friction between molecules, lowering viscosity, which is vital as it reduces the energetic requirements of the animal producing the fibre, a key factor governing the evolution of the silk production process [17]. While it is well documented that silk dopes undergo shear thinning [11,18], a feature that is shared with other liquid crystals under flow [17], this is not evidence per se that silk mesophases entail more efficient processing. However, recent use of shear-induced polarization light imaging (SIPLI) [19] demonstrated that the mechanical work input required to create fibrillar structures in silk is three orders of magnitude less compared with a non-liquid crystalline synthetic polymer melt (high-density polyethylene) [20].

Like canonical silks, non-canonical silks are produced from liquid crystalline feedstocks. In final instar honeybee larvae, silk glands contain birefringent structures called tactoids (characteristic structures formed when mesophases exist in equilibrium with isotropic phases [13]) surrounded by an isotropic fluid [21–23]. Consistent with this notion, tactoids are observed to form first at the periphery of the gland lumen, with their tips terminating at the surface of cuboidal cells where silk proteins are secreted. Later, tactoids are observed within the entire gland lumen [21,23]. In the silk glands of bumblebees and hornets, silk dope has been observed to form a pattern called ‘fibrous bars’ which we suggest is actually the product of fusion of tactoids into a cohesive mesophase spanning the silk gland. Honeybee tactoids show a regular banding pattern with periodicity of 500 nm, while the ‘width’ of the fibrous bars is 1000–1600 nm. In both cases, the banding pattern is likely to correspond to a repeating aspect of the mesophase structure such as the pitch of a chiral nematic (see §1c).
(ii) Control of solubility and aggregation

Silk must be capable of solidifying as it leaves the body of the organism, but premature solidification inside the silk gland must be avoided. As mesophases form at high mesogen concentration but still retain liquid properties, mesophase formation is probably the major mechanism by which solubility is maintained and aggregation avoided. In silkworms and spiders, solubility of silk proteins is also maintained during storage by modulation of dope pH, ionic composition and water content [24]. As silk dope flows anteriorly towards the spinneret along an increasingly narrow duct, the pH is lowered, cation concentrations increase, and in the case of Lepidopteran silks (i.e. B. mori) a sheath of sericin proteins is thought to have the effect of dehydrating the dope. Each of these changes primes the dope for aggregation and the liquid → solid transition.

In many silks of both the canonical and non-canonical types, covalent cross-linking upon extrusion, either owing to oxidative cross-linking through cysteine residues or due to enzymatic tanning, may assist aggregation and solidification [7,25]. However, covalent cross-linking is not a requisite feature of fabrication of either canonical or non-canonical silk proteins, as demonstrated by the potential for silkworm, hornet and sawfly silks to be dissolved in chaotropic solutions without reducing agents [26–28].

(iii) Formation and orientation of crystallites

In the model system of the silkworm, proteins in silk dope have a structure rich in β-turns with substantial conformational flexibility (the ‘silk I’ structure) [29]. Rheological studies combined with various spectroscopic, microscopic and scattering techniques suggest mesophase intermediates are also involved in the development of molecular and supramolecular structure in silkworm glands [14]. On the nanoscale, Rheo-IR studies have indicated that silk proteins will align in response to flow and that alignment occurs prior to silk I → silk II conversion [30]. On the microscale, confocal rheology has demonstrated the development of fibrillar structures oriented along the shearing direction [31] and on the macroscale SIPLI has shown flow-induced birefringence [20]. However, care must be taken to assign specific structures responsible for the birefringence patterns observed using visible light microscopy either in the fibre or the gland itself [16,32] as the liquid crystalline texture of the dope disappears prior to the silk I → silk II conversion. Therefore, the structural unit (mesogen) is unlikely to comprise solely β-sheet crystallites [14] (see §3a).

The process of converting silk I to the final silk II structure (extended-β-sheet crystallites surrounded by amorphous chains) is achieved by mechanically deforming the dope as it flows down the duct prior to its emergence from the spinneret. Mechanical deformation causes molecular extension of fibroin proteins due to shearing forces from friction caused at the duct wall, extensional forces due to the drawing of the material, and elongational flow. Molecular extension brings adjacent chains into close apposition, promoting dehydration, the formation of intermolecular hydrogen bonds and aggregation, and hence the silk I → silk II conversion yielding aligned β-sheet crystallites [14,33]. Thus, the directionally applied mechanical deformation both contributes to the formation of crystallite secondary structure and the orientation of crystallites with respect to the fibre axis.

Crystallite formation and orientation in non-canonical silk processing can be inferred to be quite different compared with processing of silk by silkworms and spiders. In canonical silks, β-strands within crystallites are on average parallel with the fibre axis and therefore fully extended in this direction. By contrast, protein chains in non-canonical silks present as α-helices, collagen helices and cross-β-sheet crystallites lie at an angle to the fibre axis and may be further extended by applying tension to the fibre to form extended-β-sheets [4,34]. Therefore, the crystallites are unlikely to be formed in response to molecular extension during silk processing. Consistent with this reasoning, recombinant versions of aculeate hymenopteran and sawfly silk proteins form mature secondary structures similar to crystallites in solution, without the application of mechanical force [28,35] (see §2b). In addition, as crystallites are not formed due to forces acting parallel to the fibre axis during the liquid-to-solid transition, an alternative process must direct their orientation within the non-canonical silk fibres. We will argue that liquid crystalline states are ideally placed to direct this process, though in a fundamentally different manner to what occurs in canonical silk processing.

(c) In search of a mesogen: the mechanism of mesophase formation by silk proteins

Liquid crystals (mesophases) are states of matter having the properties of both liquids and crystalline solids. For example, molecules in close proximity may share alignment (orientational order) and/or a lattice-like arrangement (positional order) while retaining the ability to flow like a liquid. The first known liquid crystals were preparations of elongated organic molecules in which mesophase formation could be induced by temperature changes (thermotropic mesophases). The key feature that causes some molecules to be mesogenic and some not is their shape. A length/width ratio (axial ratio) above approximately five is sufficient for mesogenicity, provided the molecule can be concentrated; the higher the axial ratio of a molecule, the lower the concentration required to induce a mesophase [36]. Mesophase formation is a spontaneous event that occurs because the loss of orientational entropy associated with molecular alignment is outweighed by a gain in positional entropy [37]. In the simplest case, entropic minimization results in a nematic phase where the molecules share local orientational order without having positional order; the addition of positional order yields a smectic phase. Finally, chiral molecules may form layers at a preferred angle to underlying layers, producing chiral nematic and chiral smectic phases.

Lyotropic mesophases are more complex states of matter than thermotropic mesophases and form due to the interaction of two different components. A typical example is the formation of lipid bilayers—a type of liquid crystal—as a simple consequence of mixing amphiphilic molecules such as phospholipids and water. Apart from temperature, the key variable controlling the structure of the mesophase formed is the relative concentration of the two components. Phases dominated by vesicles, micelles, hexagonal columns, lamellae, cubic and inverse phases are formed with successively increasing concentrations of phospholipid [38]. Importantly for our discussion of mesogen structure in canonical silk processing (§3a), anisotropic lyotropic structures such as hexagonal columns and lamellae are subject to the same entropic considerations discussed above for small molecules, and thus may form large-scale versions of the nematic and smectic phases.
behaviour of regenerated and recombinant silk proteins. Sequences, studies on natural silk spinning can be inferred by analysis of silk protein amino acid exact nature of mesogens in silk dope is unknown, but much globular proteins (tobacco mosaic virus (TMV), F-actin). The lagen, PBLG) or end-to-end polymerization of multiple small either through the use of intrinsically rod-like structures (col-
studied by Robinson [43]. High axial ratios can be achieved et al. example, the 1500 nm F-actin filaments studied by Suzuki [41] are orders of magnitude larger than PBLG molecules studied by Robinson [43]. High axial ratios can be achieved either through the use of intrinsically rod-like structures (collagen, PBLG) or end-to-end polymerization of multiple small globular proteins (tobacco mosaic virus (TMV), F-actin). The exact nature of mesogens in silk dope is unknown, but much can be inferred by analysis of silk protein amino acid sequences, studies on natural silk spinning in vivo, and the behaviour of regenerated and recombinant silk proteins.

2. Comparison of canonical and non-canonical silk proteins
(a) Length, architecture and repetition
Comparison of non-canonical and canonical silk protein sequences reveals marked differences in length, architecture and sequence repetition (figure 2). The main structural protein in silkworm silk is heavy-chain fibroin (H-fibroin, 350 kDa), which is linked covalently to light-chain fibroin (L-fibroin, 26 kDa) and non-covalently to the glycoprotein fibrohexamerin (P25, 30 kDa) to form a complex with stoichiometry 6:6:1 [3,44]. Orb-spider dragline silk is composed of two major proteins, MaSp1 and MaSp2, which have molecular weights in the range 250–500 kDa [3]. Thus, canonical extended-β-sheet-forming silk proteins are typically larger than 250 kDa. By contrast, non-canonical silk proteins from aculeates, sawflies and lacewings showing convergence to short sequences with high-repeat regularity. For example, aculeate silk is made up of four coiled-coil-forming proteins, silk fibroins 1–4. While the molecular weight of these proteins ranges between 29 and 45 kDa, the length of the coiled-coil domain is always approximately 30 heptads [45]. Sawfly silk is made up of three fibroins, StColLA-C, that range between 22 and 32 kDa but each contain either 78 or 79 collagen tripeptides [28]. The lacewing egg-stalk proteins MalXB1 and MalXB2 are larger and less constrained in length compared with the superhelical silk proteins, being 86 kDa (50 repetitive motifs) and 54 kDa (29 repetitive motifs), respectively [7].

Analysis of non-canonical silk protein amino acid sequences suggests they have a common elongated tertiary structure. Whereas ‘spacer’ regions are a characteristic feature of canonical silk proteins [2,24], non-canonical silk proteins usually show no (or at most one) interruptions to the register of the central repetitive domain (figure 2). For example, the coiled-coil prediction algorithm MARCOIL [46] predicts a coiled-coil domain of approximately 210 residues in aculeate silk proteins, probably uninterrupted by heptad irregularities such as stutters or stammers [45]; the 78–79 (Xaa–Yaa–Gly) repeats in sawfly collagen silk proteins occur consecutively without interruption; and the cross-β-sheet-forming MalXB1 and MalXB2 contain one and zero interruptions within their repetitive domains, respectively. This high repeat continuity combined with the inherently rod-like nature of coiled-coils and collagen helices, or the lath-like nature of β-hairpin ribbons, is a formula to produce highly elongated proteins in the folded state. We estimate that the central repetitive domain of each type of non-canonical silk protein is between 25 and 70 nm long in the folded state, and has an axial ratio between 14 and 40 (table 1). Thus, non-canonical silk proteins have all the necessary features required of a mesogen (i.e. axial ratio and varying hydrophilicity), and folded proteins are likely to form mesophases spontaneously when appropriately concentrated.

(b) Amino acid composition and protein folding
The propensity of amino acid residues towards particular secondary structures, as determined from solved protein structures [47], is the basis of secondary structure prediction algorithms such as GOR4 [48]. Residue folding propensities
are derived from the structures of globular proteins in aqueous environments, and are therefore a better guide to the structure of proteins in silk dope than in the final solid silk. Interestingly, canonical silk proteins are usually poor in the classic β-sheet-forming residues Val, Ile, Tyr, Phe, Cys and Trp. Instead, the small residues Gly, Ala and Ser are dominant. Consistent with this discrepancy, canonical silk proteins do not fold into β-sheet-rich structures spontaneously in solution. Reconstituted silkworm silk, before exposure to mechanical shear or solvents such as MeOH, exists in the β-turn-rich silk I structure [29,49]. Owing to its amino acid composition, GOR4 predicts silkworm H-fibroin to take predominantly a random coil conformation (85.5%) with minor β-sheet (9.4%) and α-helical (5.1%) components. This can primarily be explained as reflecting the primary importance of extrinsic factors, especially mechanical deformation, for the folding of canonical silk proteins into β-sheet-rich structures.

By contrast, the residues with the highest propensities to form α-helices and coiled-coils (Glu, Lys, Leu, Arg and Ala; [47,50]) collectively make up 38–64% of the honeybee fibroins AmelF1-F4 [25]. The GOR4 algorithm predicts 70–90% of each protein to fold into α-helices, and circular dichroism experiments using recombinantly expressed proteins confirm that honeybee fibroins fold into native-like coiled-coil structures in solution [51]. Similarly, sawfly collagen fibroins have primary structural features sufficient to induce the collagen structure in recombinant silk proteins in solution [28]: the high number of collagen tripeptide repeats alternating with hydrophilic linkers [24,33], is highly suited to the formation of micelle-like lyotropic mesophases. Viney [39] argued persuasively that the low level of birefringence exhibited by mesophagic silk dope is unlikely to be due to orientation birefringence resulting from the long-range orientation of polarizable bonds but instead represents form

<table>
<thead>
<tr>
<th>protein</th>
<th>predicted geometry</th>
<th>estimated length of mesogen (nm)</th>
<th>predicted threshold concentration for mesophase (v/v)%</th>
<th>observed threshold concentration for mesophase (v/v)%</th>
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<td>2.8 × 0.54</td>
<td>16.7b</td>
<td>42</td>
</tr>
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*Using average of width dimensions.

3. Multiple trajectories through liquid crystallinity to solid silk

(a) What is the structure of the canonical silk I mesogen?

In silk glands, the protein structures that constitute mesogenic units have not been clearly characterized. However, taking again the silkworm as a model, we can place various limits on what its structure might be. For example, as the H-fibroin protein makes up the majority of the silk, and the mesogen must occupy a large volume of the dope [37], it is likely that the mesogen contains all or part of the H-fibroin protein. Possibly, the mesogenic unit may correspond to the 6:6:1 complex formed by the three main silk proteins [44], to a single H-fibroin chain or part thereof, or to an end-to-end aggregation of multiple 6:6:1 complexes [39]. At the level of secondary structure, we can rule out the possibility that the mesogen consists of β-sheets, as mesophase occurs along the length of gland duct but disappears before the silk I → silk II structural transition near the silk press [13,14]. Instead, the major secondary structure present in the mesogen is likely to be the β-turn-rich silk I.

We propose the flexible and amphiphilic nature of H-fibroin in the silk I structure, consisting of long stretches of hydrophobic repeats alternating with hydrophilic linkers [24,33], is highly suited to the formation of micelle-like lyotropic mesophases. Viney [39] argued persuasively that the low level of birefringence exhibited by mesophagic silk dope is unlikely to be due to orientation birefringence resulting from the long-range orientation of polarizable bonds but instead represents form
birefringence, originating from the anisotropic distribution of two coexisting phases which are themselves isotropic. In the case of H-fibroin, these two phases may be best described as: (i) a hydrophobic phase consisting of the dominant repeats of the silk protein; and (ii) a hydrophilic phase consisting of water, ions and the hydrophilic linkers of the silk protein. Anisotropy may arise if conditions inherently favour anisotropic lyotropic structures such as hexagonal columns, or alternatively if isotropic structures such as micelles aggregate through head-to-tail polymerization as has been observed under conditions of shear [53]. Either way, the mesogen probably resembles micelles observed in native and reconstituted silk [33,53] with the single added feature of elongation (figure 3). Analysis of silk protein sequences and physical characterization of silk dope and fibres suggests this model may be generalized to other groups that produce silk with extended-$\beta$-sheet crystallites, such as spiders [2] and raspy crickets [5].

(b) Multi-purpose mesogenic and crystalline protein structures in non-canonical silks

In contrast to the mechanism by which canonical silk proteins form mesogens, the best candidate for mesogens formed by non-canonical silk proteins corresponds to a single-coiled-coil superhelix, collagen triple helix or cross-$\beta$-sheet ribbon (figure 3). Non-canonical silk proteins, with a central semi-rigid superhelix and more flexible domains at each terminal, appear ideally suited to form liquid crystalline mesophases. Applying standard molecular dimensions to silk proteins produced by aculeates, sawflies and mother lacewings, the rigid portion of each molecule is predicted to have an axial ratio between 14 and 40 (table 1). According to Onsager’s relationship [37], the corresponding concentration thresholds for mesophase formation fall between 19 and 50% of total volume (though experimentally measured thresholds where they are available are substantially lower). As silk proteins are known to accumulate to 30–40% of dry weight in some silk glands [54], it is reasonable to suppose non-canonical silk proteins reach these concentration thresholds.

The major conclusion of our analysis of non-canonical silk proteins (§2) is that the convergently evolved architecture of non-canonical silk proteins probably arose due to selection pressure for proteins that can act both as mesogens in the liquid phase and crystallites in the solid phase, without a major structural transition. Non-canonical silk processing thus contrasts with canonical silk processing, in which...
proteins are specialized to transition between discrete mesogenic (silk I) and crystalline (silk II) forms. As each processing mode entails numerous adaptations in protein amino acid sequence and gland structure, it is likely that effective ‘fitness troughs’ [8] prevent their interconversion during evolution, and that they therefore represent distinct fitness peaks in the adaptive landscape of silk production.

(c) Towards an understanding of non-canonical silk processing

The potential of non-canonical silk proteins to act both as mesogens and crystallites suggests a unique silk processing pathway. It has long been recognized that materials such as insect cuticles, plant cell walls and interstitial collagens can be considered solid analogues of mesophases [38]. The same is true of non-canonical silks, but we propose it is true in the more direct sense that they are the actual product of solidifying liquid crystalline phases. For example, in aculeate and sawfly silk, protein superhelices are arranged roughly parallel to the fibre axis, but without positional orientation with respect to adjacent superhelices [4], i.e. they are solid analogues of nematic mesophases.

Based on the analysis above, we are now in a position to propose a tentative model for the process of silk spinning using non-canonical silk proteins (figure 3). Firstly, proteins folded into rod- or lath-shaped molecules accumulate in silk gland lumen. Above a critical concentration, the molecules transition into a mesophase, producing local alignment of molecules. Near the spinneret where the gland walls are close together, the average orientation of the mesogens undergoes flow-induced alignment in the direction of the spinneret, which is parallel to the eventual fibre axis. Non-canonical silk proteins are positioned closely together in mesophase, interacting via non-covalent bonds with water and solute molecules, and with adjacent proteins. As the lumen is drawn from the gland to the external air, sufficient bonds are present between adjacent proteins to overcome capillary break-up. This mechanism is consistent with the wetting properties of fibre bundles where the spreading parameter is negative, meaning bundles of fibres are naturally preferred [55]. Hence, the fibre is stable over the timescale required for it to dehydrate and for protein–protein bonds to replace protein–water bonds. After solidification, the fibre is complete and has molecular orientation that is continuous with the orientation in the preceding liquid crystal phase.

We note that this kind of fabrication is similar to the fabrication of materials such as dogfish egg-sacks [56] and mantis ootheca [57]. In both cases, concentrated protein solutions progress through distinct mesophases before solidifying as analogues of lamellar and smectic phases. Moreover, the proteins present in these materials show numerous similarities at the amino acid sequence level to non-canonical silk proteins. However, it has not previously been appreciated that fabrication of this type is a feasible route for producing solid, cylindrical, micrometre-scale fibres on demand.

(d) New artificial protein materials through biomimetic self-assembly of silk mesogens

The ability to produce recombinant versions of non-canonical silk proteins has generated interest in their use for the creation of artificial protein materials [28,35,58–60]. However, attempts to date make use of non-natural processing steps, such as solubilization in hexafluoroacetone [58] or detergent micelles [35,51,60], and solidification using methanol baths [35,60]. It is not clear what kind of liquid crystal phenomena, if any, occur during these types of material fabrication. Interestingly, precedent does exist for the formation of artificial fibres using folded, rod-shaped proteins in mesophase.

Over the past two decades, systems have been developed in which the mesogenic properties of elongated virus capsules direct the self-assembly of hierarchically structured materials [61]. For example, an engineered ZnS-binding TMV was induced by Lee and co-workers to form smectic phases that were then solidified, allowing the production of a highly ordered material with regularly spaced metallic particles [62]. Further advantages may be gained on these already sophisticated fabrication systems if it is possible to harness the natural and straightforward mesogenic behaviour that we suggest characterizes non-canonical silk proteins. For example, aculeate silk proteins are around 30 nm long (cf. 300 nm for the TMV) and hence might be used to introduce structure at finer scales compared with virion mesogens. In addition, non-canonical silk proteins are easily engineered, express at high levels and are easily refolded into native-like conformations [28,35,60]. However, the greatest advantage of non-canonical silk proteins over alternative protein mesogens, such as virus particles, is that liquid crystalline processing into solid materials is their natural function. Accordingly, their primary sequences are likely to contain numerous adaptations predisposing them to act as mesogens in the liquid state, and strong and flexible structural proteins in the solid state.

4. Conclusion

Silk proteins that form crystallites with structures other than extended-β-sheets have convergently evolved features that distinguish them from canonical silk proteins made by silkworms and spiders. Non-canonical silk proteins are characterized by relatively small size (less than 86 kDa) and high-repeat continuity within the central repetitive domain. This primary structure yields a rod-like or lath-like tertiary structure 23–70 nm in length, with more flexible domains at each end. Importantly, this structure is likely to be formed spontaneously in solution without reliance on extrinsic factors.

We propose the evolutionary convergence of this rod/lath-like structure in non-canonical silk proteins is the result of selection for proteins that can act efficiently both as mesogens in the liquid state and crystallites in the solid state, without an intervening transition in secondary or tertiary structure. This lies in stark contrast to canonical silk proteins, which are highly specialized to both forms, mesogens and crystallites, but which undergo extensive changes at the level of protein structure to transition between the two forms. We propose a simple qualitative model of silk fabrication using non-canonical silk proteins that further suggests the molecular orientation in the solid silk is continuous with molecular orientation in the liquid crystal state, i.e. that non-canonical silks are fabricated by solidification of mesophases. Recombinant non-canonical silk proteins from aculeates and sawflies, and structural analogues of lacewing egg-stalk proteins, have been used to make biomimetic materials but processing so far only poorly mimics natural spinning
and hence is unlikely to capture the full potential of these proteins for assembling hierarchical structures. We conclude that further investigation of liquid crystalline mesophases formed by recombinant non-canonical silk proteins is likely to yield further advances towards the creation of sophisticated protein materials.

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